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Mobility of O_2^- ions in supercritical Ar: Experiment and molecular dynamics simulations

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1. Introduction

Negative ions in supercritical gases are an interesting subject for several reasons. The formation of anions as a consequence of low-energy electron impact to a molecule has important consequences in the chemistry of atmosphere, of discharges, and in electro- and biochemistry. Electron attachment and detachment processes, leading to the formation and decay of anions are strongly influenced by the environment [1]. Upon creation and stabilization, they interact with the environment leading to states that cannot be adiabatically obtained by a simple addition of the ion to the surrounding system [2,3].

The investigation of the transport properties of ions in a dense supercritical gas gives important pieces of information about their structure and interaction with the host [4] and bridge the gap between dilute gases [5] and liquids [6].

Anions are difficult to produce even in a dense environment, with the noticeable exception of oxygen and SF_6 that are very electronegative. In particular, O_2 is a ubiquitous species, present as impurity even in the best purified rare gas. Resonant electron attachment [7], followed by stabilization induced by three-body collisions [8], produces stable O_2^- ions that can be easily drifted

ABSTRACT

A new analysis and new molecular dynamics (MD) simulations of the measurements of the mobility μ_i of O_2^- ions in dense supercritical Ar gas are reported. μ_i shows a marked dependence on the distance from the critical temperature T_c . A mobility dip appears as a function of the gas density and its maximum value occurs below the critical density. The locus of points of maximum mobility dip in the *P*–*T* plane appears on the extrapolation of the coexistence curve into the single-phase region. MD simulations quantitatively reproduce the mobility dip near T_c .

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through the gas under the action of an externally applied electric field.

The O_2^- ion couples with the gas by means of electrostriction [9]. Its strong field polarizes the surrounding atoms which are attracted toward the ion that becomes solvated in a cluster. Electrostriction thus induces rapidly varying density and viscosity profiles around the ion [4,10,11] that affect the ion mobility μ_i . In these conditions, the ion transport is actually determined by the hydrodynamic interaction of the large solvation structure surrounding the ion with the gas and is quite insensitive to the ionic species.

Whereas μ_i in He at T = 77.2 K [12] does not show any particular features as a function of the gas density N up to moderate values, it shows a smooth change of behavior when N is varied across the critical density N_c in Ne gas 0.6 K above the critical temperature T_c [4]. The dependence of μ_i on N for $N > N_c$ has been described in terms of the Stokes' hydrodynamic formula relating μ_i to ion radius R and fluid viscosity η : $\mu_i = e/6\pi\eta R$, in which η is evaluated for the density of the maximum of the density profile induced by electrostriction [13]. Unfortunately, the effective radius R cannot be determined independently and must be deduced by fitting the data to the Stokes' formula.

Further measurements in near critical Ar gas [14] have confirmed that the coupling of the ion motion with the critical fluctuations via electrostriction is responsible for the change of behavior of μ_i near the critical point. The behavior close to T_c has been explained by modifying the Stokes's formula in order

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to account for the fact that the large gas compressibility forces a layer of correlated fluid to stick around the ion. In order to test this interpretation at microscopic level, molecular dynamics (MD) simulations are performed.

2. Experimental results and MD simulations

The experimental results of the measurements of the O_2^- ion density-normalized mobility $\mu_i N$ in dense supercritical Ar for T = 157, 154, and 151.5 K are shown in Fig. 1. Details on the experiment and on the previous analysis are found in literature [14]. For a density $N_m < N_c = 8.08 \text{ nm}^{-3}\mu_i$ shows a dip whose strength increases when $T \rightarrow T_c = 150.86$ K. The solid line through the data on the isotherm closest to T_c is obtained by using the Stokes' formula with an ion radius *R* that is enhanced with respect to the radius in absence of electrostriction and criticality by an amount that depends on the local value of the fluid correlation length ξ [14]. The good agreement confirms the picture of a correlated fluid layer dragged by the ion because of electrostriction. For $N < 4 \text{ nm}^{-3}$ the hydrodynamic description fails to reproduce the data because the ion transport leaves the hydrodynamic regime to enter the kinetic one.

A new analysis of the data reveals that the locus of the maximum mobility dip lies on the extrapolation of the coexistence curve into the single-phase region, as shown in Fig. 2. The same occurs for ³He+ ions in ³He near the critical point for $T_c \approx 3.3$ K [15].The gas appears reminiscent of the nearby coexistence curve even for $T > T_c$. Droplets of superheated liquid are formed in the gas because of critical fluctuations. Electrostriction makes the local pressure around the ion rise through the extrapolation of the coexistence line into the region of superheated liquid that the strong ion field stabilizes thermodynamically. This situation is also similar to the case of ⁴He+ ions near the melting transition in which the λ -line is extrapolated into the region of supercooled, electrostrictively stabilized liquid surrounding the ion [16].

To get a microscopic insight of the ionic transport process, classical MD simulations have been carried out in which the Newtonian equations of motion are numerically integrated for suitable interaction potentials by using freely available Fortran codes [17]. A system of 100 Ar atoms plus 1 O_2^- ion has been simulated as a function of N for T = 151.5 K. The Ar–Ar interaction potential is found in literature [18]. The dominant long-range part of the O_2^- –Ar potential can be calculated by the knowledge of the polarizabilities of O_2^- and Ar and of the quadrupole moment of Ar [19]. The repulsive part



 $N (nm^{-3})$

6

8

10

12

14

 $\mu_{N}[10^{21} (\mathrm{V \,m\,s})^{-1})]$

0 0

2

4



Fig. 2. Locus of points of minimum O_2^- ion mobility. Solid line: coexistence curve of Ar [21]. Dotted line: extrapolation of the coexistence curve into the single-phase region.

of the O_2^- -Ar potential has been modeled as the interaction of the neutral species with an infinite wall endowed with a local charge density equal to the electronic charge density of the ion [20].

The equations of motion are integrated by using the Verlet algorithm [17] in time steps of $\tau = 100\tau_i = 2.42 \times 10^{-15}$ s, where $\tau_i = 4\pi\epsilon_0 a_0 \hbar/e^2$ is the timescale in atomic units (a.u.). After an initial equilibration time, the system evolves freely. The coordinates of all particles are recorded every time interval $t_n = (500\tau)n$, $n = 1, \ldots, n_f$, spanning a total simulation time ≈ 2.4 ns for $n_f = 2000$. In Fig. 3 we report a typical example of the mean square displacement $\langle r^2 \rangle$ of the ion and of Ar for N = 6.5 nm⁻³ and T = 151.5 K. For Ar, it is less noisy because of the higher statistical significance due to the averaging over 100 atoms whereas the ion displacement is relative to the only one ion present in the simulation. As expected, the ion is less mobile than the neutral atoms because the enhancement of its interaction with atoms due to its electric field.

The diffusion coefficient *D* is obtained as $D = \lim_{\Delta t \to \infty} \langle r^2(\Delta t) \rangle / 6\Delta t$. The poor statistics related to the ion may result in a not complete evolution of the mean square displacement of the ion so that a reliable diffusion coefficient can be produced. In order to improve the statistical significance of the data, the literature suggestion [17] has been followed according to which a running average of the diffusion coefficient D_{ra} is first calculated by evaluating $\langle r^2 \rangle$ for each value of the sampling window Δt by moving its time origin across the whole simulation, and averaging over the partial results before averaging over the



Fig. 3. Mean square displacement $\langle r^2 \rangle$ vs. *t* for N = 6.5 nm⁻³ and T = 151.5 K. Thin line: Ar. Thick line: O_2^- .



Fig. 4. Running average of the diffusion coefficient D_{ra} as a function of the sampling window length Δt for T = 151.5 K and N = 6.5 nm⁻³.

sampling window length Δt . In Fig. 4 a typical result for the running average of D_{ra} is plotted as a function of the sampling window Δt . The error bars are the results of averaging over the different time origins.

For practical purposes, the limit in the definition of *D* is replaced by the mean of the running average D_{ra} . Finally, the Nernst–Townsend–Einstein relationship $D/\mu_i = k_B T/e$ is used to calculate the ion mobility because no electric field, though small, is present in the simulation. Actually, its presence would have increased the temperature of the gas whereas no numerical thermostat has been implemented in the simulation code.

In Fig. 5 the MD results for T = 151.5 K are compared with the experimental data. The quantitative agreement between data and simulation is quite good.

The simulated μ_i shows a dip at the same density of the experiment. The density width of the mobility dip calculated in the simulations is larger than the experimental one. It is believed that this effect is due to the fact that the temperature, during simulation, is allowed to substantially fluctuate in order for energy to be constant. Close to the critical point, even small temperature fluctuations result in large density fluctuations. It has to be noted that, in spite of the poor statistical quality of the simulation data, nonetheless, the simulation results convincingly aim at proving that electrostriction strongly influences the ion behavior.

Disappointingly, however, MD simulations do not shed light on the physical reasons of their success. Actually, MD calculations also provide the ion-atom pair distribution function g(r) that gives



Fig. 5. Comparison between experiment (closed points) and MD simulations (open points). The error bars are not shown for graphical clarity. Solid lines: eyeguides.



Fig. 6. N_{exc} vs. *R* for $N = 6.5 \text{ nm}^{-3}$ and T = 151.5 K (left- and bottom-axes). N_{exc} vs. *N* for T = 151.5 K. Squares: 1st solvation shell. Circles: 2nd solvation shell (top- and right-axes).

important information on the structure of the ion environment. In particular, the excess number N_{exc} of atoms surrounding the ion can be calculated. In Fig. 6 N_{exc} is reported as a function of the distance R from the ion for N close to the density of the mobility minimum. The formation of the first solvation shell is clearly indicated by the plateau around $r \approx 4$ Å, while a diffuse second solvation shell is observed for $r \approx 9$ Å. In Fig. 6, the number of excess atoms contained in the 1st and 2nd solvation shells is also shown as a function of N. Interestingly enough, N_{exc} in the 2nd solvation shell is strongly peaked for $N < N_c$, thus suggesting that electrostriction efficiently modifies the ion environment when the unperturbed gas density is well below N_c . By contrast, nothing anomalous shows up in N_{exc} around the density of the mobility dip. So, there is no obvious correlation between the excess atom number in the solvation shells and the mobility dip. We think that carrying out more MD simulations will improve their statistical significance and shed light on the physical picture beneath the experiment.

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